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LETTER TO THE EDITOR

Mott *g*-ratios in $Rb_x(NH_3)_{1-x}$ and oxidation state of rubidium compounds from XAS

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Abstract. The x-ray absorption spectra (xAs) of Rb metal, $Rb_x(NH_2)_{1-x}$, 2H-NbSe₂Rb_{0.28} and RbBr near the Rb K-edge have been used to ascertain that the oxidation state V of rubidium dissolved in ammonia and intercalated in the layer compound is in the range 0 < V < 1. The observed edge shifts with temperature for semimetals are explained in terms of the population of band states, and the ratio of the density states near the mobility edge over that calculated for a free electron model, i.e. the Mott ratio g, is ascertained using a semiempirical relation developed for the x-ray absorbance from 1s levels to empty states near the mobility edge.

XAS is traditionally divided into an XAES region where the ejected electron transitions to bound states are observed (the so-called absorption edge) and an EXAFS region (10^2 to 10^3 eV above the latter, the so-called extended x-ray absorption fine structure region) where transitions to free electron states are observed (Kronig 1932, Hartree *et al* 1934, Stern *et al* 1975). The XAES region contains information on the oxidation state of the absorber and recent semiempirical relations have shown that the edge position increases linearly with say the valence of Mn, Fe, Mo, Rb and Sr (Cramer *et al* 1976, J A Kirby unpublished, Acrivos *et al* 1980). Also transitions to bound states of the absorber can be resolved, e.g.

$$(Z^{\nu}) (1s)^{2} \dots : {}^{2S+1}L_{J} \to (Z^{\nu}) (1s) \dots nl : {}^{2S+1}(L \pm 1)_{J+1}$$
(1)

where ... represents other core electrons not directly involved in the transition and the *nl* states are members of a Rydberg series for a given angular momentum *l* including the exciton states. Parratt and others (Parratt 1939, 1959, Cauchois and Mott 1949) have related the spacings between the series of transitions (1) with those reported in the literature for the element with atomic number Z + 1 and a closed K shell, i.e.

$$((Z+1)^{\nu})(1s)^{2}\dots n'l': {}^{2S'+1}L'_{J'} \to ((Z+1)^{\nu})(1s)^{2}\dots nl: {}^{2S'+1}L'_{J''}$$
(2)

where $S' = S \pm \frac{1}{2}$ and this valuable approximation is correct if exchange interactions between the K electrons and the other core electrons are negligible. In this work we make use of the XAES data including the edge position and the transitions (1) in order to ascertain the valence of rubidium in the metal-ammonia solutions (MAS) and in intercalated compounds for the purpose of determining the metallic nature of the system.

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The XAES data for Rb metal, RbBr, RbN₃and Rb MAS and for 2H-NbSe₂Rb_{0.28} were reported previously (Acrivos *et al* 1980, Bourdillon *et al* 1979). Here figure 1 shows that the edge inflection point E_i moves to higher energy by 4.5 eV as the valence increases from 0 in Rb metal to +1 in RbBr and RbN₃, displaying an intermediate value for Rb MAS. The edge positions for the intercalated compounds were not determined with the same standards (Bourdillon *et al* 1979); however, the most useful observations are

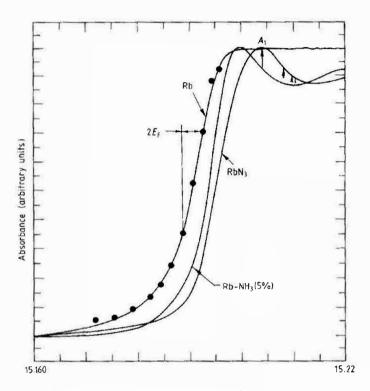


Figure 1. XAEs for different Rb compounds. A_m^0 evaluated using equation (4) with $E_F = 1.79 \text{ eV}$ is indicated by dots.

related to transitions (1), i.e.

 $(Rb^{+1}) (1s)^2 \dots : {}^{1}S_0 \to (Rb^{+1}) (1s) \dots np: {}^{1}P_1$ (1')

where $n \ge 5$. The spacing in the Rydberg series n = 5, 6, 7, 8 can be estimated from transitions (2) in the literature (Moore 1952) for Sr^{+1} , i.e.

$$(Sr^{+1}) (1s)^2 \dots 5s: {}^{2}S_{1/2} \to (Sr^{+1}) (1s)^2 \dots np: {}^{2}P_{1/2,3/2}$$
 (2')

which are indicated by arrows in figure 1 for RbN₃.

Now the changes in absorbance against T for MAS have to be explained. A semiempirical analysis shows how the variation in the density of states may be extracted from the temperature dependence of the absorbance as follows. In non-metals (NM) the transitions to a continuum of free states from a 1s level give rise to an infinite sum of

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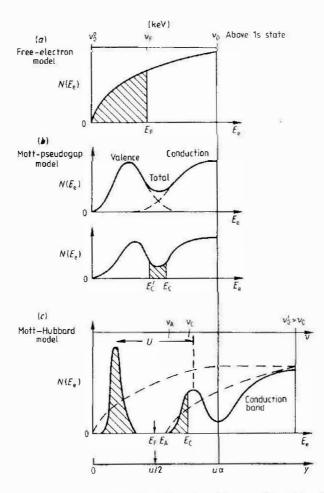


Figure 2. Density of states for different models according to Mott (1974).

Lorentzian curves which start at the edge $E_0 = h\nu_0$ in figure 2, i.e. the absorbance is (Richtmyer *et al* 1934)

$$A_{nm}(\nu) = D_{\nu}(\pi/2 + \theta_{nm}) \tag{3}$$

shown in figure 3. Here $-\pi/2 \leq \theta_{nm} = \tan^{-1}[2\pi T_{\nu}(\nu - \nu_0)] \leq \pi/2$ and $D_{\nu'\nu} = D_{\nu}$ is a function of the transition probability and the joint density of states and T_{ν} is a lifetime depending on the initial and final state widths. Then the XAES for Rb⁺ in figure 1 are explained as a series of Rydberg transitions (1) (each described by a single Lorentzian which decreases in intensity by a factor greater than two from the previous one in the series) plus (3).

In alkali metals (or alkaline earth metals) the conduction electrons occupy band states and these affect both transitions (1) and (2) as follows. The band states are occupied up to the Fermi level $E_F = k_F^2/2m = h(\nu_F - \nu_0^0)$ where ν_0^0 is the bottom of the conduction band and ν_F is the Fermi level in figure 2(*a*). However, ν_F may be in the vicinity of the final states in transition (1). In this case mixing of localised and extended states can and does occur (Fano 1961), leading to occupancy of the final states in transition (1) and to the absence of exciton states. The overall absorbance is then written:

$$A(\nu) = A_{1}(\nu) + A_{nm}(\nu)$$
⁽⁴⁾

where A_{nm} is given by equation (3), and in order to ascertain which of the different models shown in figure 2 describes the MAS correctly we write the absorbance below the

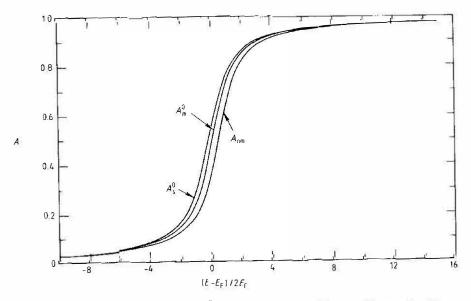


Figure 3. Normalised absorbance Λ^0 against $(E - E_1)/2e_F$ with $\alpha, u \approx 1$ in equation (4).

edge separately:

$$A_{1} = D_{\nu} \int_{\nu_{0}}^{\nu_{0}} \frac{2\pi T_{\nu} d\nu' (1-f)}{1+4\pi^{2} (\nu'-\nu) T_{\nu}^{2}} N(\nu') / N(\nu_{0}).$$

f is the Fermi distribution function for the occupation of a level in figure 2, and the density of states $N(\nu')$ has been normalised to the value at ν_0 for the metal and $D_{\nu'\nu}$ has been replaced by D_{ν} . For a spherical Fermi surface the density of states near the bottom of the conduction band varies as the square root of the kinetic energy, i.e.

$$N(\nu')/N(\nu_0) = \left[(\nu' - \nu_0^0) / (\nu_0 - \nu_0^0) \right]^{1/2}$$

and as exciton states appear variations in A_1 are expected as one goes from a metal to an insulator.

The physical significance of A_1 may be ascertained by introducing dimensionless variables y in units of $2E_F$ in equation $(4)(y = y(v) \equiv uh(v - v_0^L)/2E_F)$ as shown in figure 2, where the uncertainty principle requires that $T_v E_F \ge \hbar/2$ and defines the parameter $u \equiv 2T_v E_F/\hbar \ge 1$. Then

where y' = x - i + y and the edge will vary as a function of f as follows:

(a) For semiconductors near 0 K, f = 0 in equation (4), and the integral is evaluated using relation 2.225 of Gradshteyn and Ryzhik (1965) to give the semiconductor edge absorbance, A_s^0 , plotted in figure 3. However, as T increases $(1 - f) \le 1$ in equation (4) causes the edge to move to higher energy as shown in figure 4 for Rb-NH₃, which is discussed below.

(b) For metals when $E_F \gg k_B T$, f = 1 in (4) for 0 < y' < u/2 and f = 0 when y > u/2, giving rise to the metallic edge absorbance, A_m^0 , plotted in figure 3. The calculated points A_m^0 have been plotted over the experimental Rb metal edge in figure 1 using $E_F(Rb) = 1.79$ eV to show the type of agreement obtained.

(c) For semimetals the absorbance is somewhere between A_s^0 and A_m^0 above. At finite temperatures, if E_g is an activation energy from a valence to a conduction band, the absorbance is written by adding a term to A_m^0 , i.e.

$$A_{s} = A_{m}^{0} + \int_{0}^{u/2} (1 - f) \, dA^{0} \exp(-\beta E_{g})$$
⁽⁵⁾

which can be integrated by parts to obtain the semiconductor edge absorbance A_s , at finite temperature ($\beta = 1/k_BT$):

$$A_{s} = A_{s}^{0} - \left[A_{s}^{0} - A_{m}^{0} + 2\beta E_{F} \int_{0}^{\omega^{2}} A^{0} f(1 - f) \, dy'\right] \exp(-\beta E_{g}).$$
(6)

For a semiconductor f(1 - f) is negligible and in a first approximation the last term in equation (6) is dropped, i.e.

$$A_{\rm s}^{(1)}(y,T) = A_{\rm s}^{0}[1 - r \exp(-\beta E_{\rm g})]$$
⁽⁷⁾

where $r = (A_s^0 - A_m^0)/A_s^0$ is the fractional reduction of the absorbance due to the population of states between the bottom of the conduction band and ν'_0 and the magnitude of r evaluated from equation (4) is ~0.2 near ν_F . The edge absorbance changes with β as

$$(\partial \ln A_{\rm s}^{(1)}/\partial \beta)_{\rm y} = E_{\rm g}(A_{\rm s}^0 - A_{\rm s}^{(1)})/A_{\rm s}^{(1)}$$
(8)

and the magnitude of y near the edge inflection point increases with T, because in equation (7)

$$A_{\rm s}^{(1)}(y, T)/A_{\rm s}^{0}(y, 0\,{\rm K}) = 1 - r\,\exp(-\beta E_{\rm g}) < 1 \tag{7'}$$

causing the shifts observed in figure 4. When $T = -56^{\circ}$ to -10° C for Rb-NH₃ (5 MPM) in figure 4 the slope $(\partial \ln A/\partial \beta)_y = 4 \times 10^{-3} \text{ eV}$ to $2 \times 10^{-2} \text{ eV}$ gives an energy $E_g \ge 0.26 \text{ eV}$. This is unrealistically high, indicating that the simple band model in figure 2(*a*) does not apply to MAS near the metal-insulator transition. Other band models must be considered.

(d) Mott has described various cases for the calculation of the density of states in disordered materials as shown in figure 2. The conductivity in MAS has been described qualitatively by $N(E_c)$ forming two overlapping bands where localised states are occupied up to say $E_{c'}$ in figure 2(b) (Mott 1974). Then the absorbance must be written with a correction for the density of states, i.e.

$$A_{g}(y) = A_{1g}(y) + A_{nm}(y)$$
(9)

where

$$A_{\lg}(y) = \int_{y_{\kappa'}}^{u\alpha} g(\nu') \, \mathrm{d}A^{\vartheta}$$

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where $y_{c'} = E_{c'}/2E_F$ and $g(v') = N(v')/N(v')_{\text{freelectrons}}$ is the ratio of the density of states relative to that for a free electron gas, i.e. $g_c < 1$ near $\nu = \nu_c$ is the Mott ratio and $g \rightarrow 1$ as $\nu \rightarrow \nu_0$. Integration by parts then gives

$$A_{g}(y) = A_{s}^{0} - g_{c}(A_{s}^{0} - A_{m}^{0}) - \int_{y_{c'}}^{u\alpha} A^{0} dg \qquad (9')$$

and the absorbance correct to first order, neglecting the last term in equation (9'), is

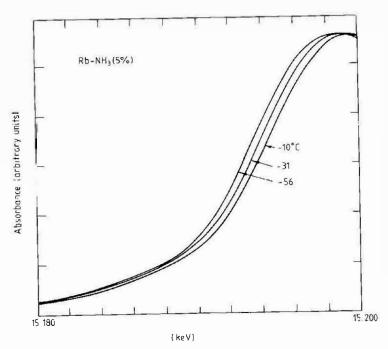


Figure 4. XAES of Rb-NH₃, 5 MPM at different *T*. Here A = 83.1; 78.2 and 67 are the averages of different measurements near 15.193 keV at -56, -31 and -10° C respectively.

written in a form similar to equation (7), i.e.

$$A_{g}^{(1)}(y) = A_{g}^{0}(y) \left(1 - g_{c}r\right)$$
⁽¹⁰⁾

where E_g in equation (7) has been replaced in equation (10) by $-(\partial \ln g_c/\partial \beta)$, which is not constant with β (i.e. $g \to 0$ as $\beta \to \infty$). We apply equation (10) to the data in figure 4 (5 MPM Rb-NH₃ with r = 0.2). Then, $g_c(-31^\circ\text{C}) - g_c(-56^\circ\text{C}) = 0.3$ and $g_c(-10^\circ\text{C}) - g_c(-56^\circ\text{C}) = 0.9$ support a density of states shape given by figure 2(b). Here $g_c(-56^\circ\text{C}) \le 0.2$ was evaluated from conductivity data (Sharp *et al* 1971).

Other models which predict a density of states which changes with temperature near the mobility edge, such as the Mott-Hubbard model in figure 2(c), would also explain the absorbance near ν_F or ν_c . Here, if the density of states decreases for E_c above E_c as in figure 2(c), the absorbance should show additional structure above the onset of the continuum of states. Also, when $\nu'_0 \ge \nu_0$ in figure 2(c) and the edge is identified near ν_c , the decrease in the density of states as $E_c > E_c$ would affect the EXAFS relation. Here

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the amplitudes measured relative to $A(\nu_c)$ would give a smaller number of scatterers about the absorber. This has been observed in M-NH₃ (Acrivos *et al* 1980) but a qualitative interpretation of results may be accomplished with the $N(E_e)$ given in either figure 2(b) or 2(c).

In summary, relations (4) are to be compared with the data in figures 1 and 4 as follows:

(i) The value $E_{\rm F}({\rm Rb}) = 1.79 \,{\rm eV}$ used in relation (4) fits the XAES date for Rb metal in figure 1 in the region where D_{ν} is not expected to change very fast. Here it is interesting to note that the edge sharpness depends on $E_{\rm F}$ alone if the uncertainty principle gives $T_{\nu}E_{\rm F} \ge \hbar/2$, which assumes that the width of the final states is greater than that for the 1s initial state.

(ii) E_F for Rb_{0.05} (NH₃)_{0.95} has been estimated to be 0.4 eV (Thompson 1976, 1977, Sharp *et al* 1971), but the T = 0 absorbance A_s^0 evaluated using equation (4) is much sharper than the observed edge in figure 1. This is probably due to temperature effects.

(iii) The Rb edge in 2H-NbSe₂Rb_{0.28} (Bourdillon *et al* 1979) shows exciton structure similar to A of MAS not Rb metal in figure 1 suggesting that the intercalated layer is not metallic. The reason for this may arise from two possibilities: (a) the Rb atoms have been separated in the intercalated layer causing a true Mott transition (Mott 1974); or (b) charge transfer may have occurred from the Rb layer into the NbSe₂ layer. The accurate edge position and its *T* dependence is necessary to determine which of these two possibilities is correct.

In conclusion, it has been shown that XAES data can be used to ascertain the nature of metal-insulator transitions in semimetals such as M-NH₃ and intercalated layer materials where it is not possible to use other conventional methods. This is of some importance in order to ascertain the metallic properties of intercalated compounds and M-NH₃ where the metallic properties of the constituent layers are different and where other measurements, e.g. Knight shift data, give only an average g-ratio over all the collisions occurring in 10^{-10} s (Acrivos and Mott 1971). Also, the Mott ratio which is proportional to the conductivity squared can be evaluated from experimental data only for values above minimum metallic conductivity of $200 (\Omega \text{ cm})^{-1}$ but g_c can be evaluated from equation (10) for any metal concentrations and serves to show for the first time how thermal disorder leads to increase the overlap between two Hubbard bands.

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Appendix: glossary of terms used

- metal-ammonia solutions MAS
- x-ray absorption spectroscopy XAS
- x-ray edge absorption spectroscopy XAES
- Vvalence of absorber

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- Ζ atomic number of absorber
- absorbance of x-rays by a given material: A
 - for non-metals A_{nm}
 - for semimetals A_{ϵ}
 - for metals A_{m}
 - for materials with overlapping bands
 - $egin{array}{c} A_{\mathfrak{g}} \ A^0 \end{array}$ for material near 0 K
 - first-order approximation (neglecting $(1 f)f \ll 1$ A(1) and/or $dg \approx 0$)
 - absorbance below the edge E_0 . $A_1 = A - A_{nm}$

Frequency of x-ray photons measured from the 1s level of absorber:

- v_0^0 to the bottom of the conduction band
- to the Fermi level $v_{\rm F}$
- to the edge of the continuum of free states in non-metals ν_0 $(E_0 = h v_0)$
- $y = y(v) = h(v v_0^0)/2E_F u^{-1}$: dimensionless parameter which measures the photon energy from the bottom of the conduction band in units of a resonance line half-width at half height $(\hbar/T_{\nu} = 2E_{\rm F}/u)$.
- density of states near E = hv above the 1s level of the absorber $N(\nu)$
- relative intensity of resonance line near $E = h\nu$ D_{ν}
- activation energy in simple semimetal $2E_{e}$
- Mott ratio g measured near the mobility edge. gc